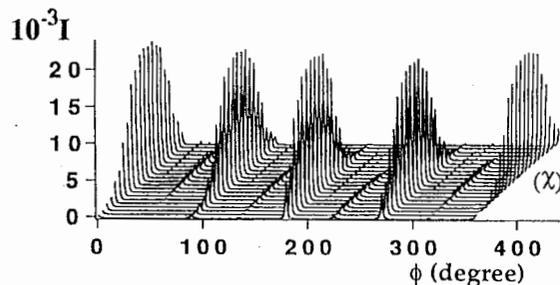


## 13-Defects, Microstructures and Textures

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X-ray  $\chi$ - $\phi$  scans of the (103/013) reflexion of YBCO film deposited on (001)MgO. The thickness of the film was 150 nm.

The studied films had a relatively constant critical temperature  $T_c$ , however a drastic reduction in the critical current density  $J_c$  was observed (from  $5.10^5$  A/cm<sup>2</sup> to  $3.10^4$  A/cm<sup>2</sup>) as the mixture ratio  $c_{L45}/c_{L0}$  increased (from 1% to 6%). This fact confirmed the drastic decreasing effect of highly misoriented grains on  $J_c$  (Dimos et al., Phys. Rev. Lett., 1988, 61, 219-222)

In conclusion the X-ray pole figure analysis is not only an effective way to study the texture of YBCO films, but also an excellent way to quantify the lattice alignment between the film and the substrate.

The authors wish to thank Alcatel-Alsthom Research for providing the samples and for critical current measurements.

**PS-13.03.12** A STUDY OF THE SURFACE DAMAGE ON A GaAs WAFER BY X-RAY DOUBLE CRYSTAL DIFFRACTION. By Ge Zhongjiu\*, Cao Wanghe, Li Mei and Liu Weina, Changchun Institute of Physics, Academia Sinica, China.

It is known from X-ray dynamical diffraction theory that the full width at half maximum (FWHM) of the rocking curve of a reflection from a perfect crystal is very narrow, about a few seconds of arc. But the surface damage and the defects in a crystal lead to a lower crystal perfection. The effects of dynamical and kinematical diffraction overlap, and the rocking curve FWHM increases. Patel suggested (J.R. Patel *et al.*, *Acta Met.*, 1962, 10, 759) that the broadening of the FWHM from the surface damage of the wafer is larger than that due to dislocations. Therefore, the depth of the surface damage layer can be detected by measurement of the FWHM at different depth from the surface of a wafer.

In this work, the surface damage on a GaAs wafer has been studied by measuring the FWHM of an X-ray double crystal diffraction rocking curve in the non-parallel (+, -) setting. Seven polished GaAs wafers have been etched to a depth of about 30  $\mu$ m. The FWHMs before and after etching are shown in Table 1. It is obvious that the differences in FWHM for samples No 1, 3, 4, 5, are slight, but the decrease in FWHM after etching is of 2 to 6 seconds for samples 2, 6 and 7. From this, we can conclude that there is a damage layer on these three samples. Fig. 1 shows a double crystal reflection topograph of a polished (001) GaAs wafer. In the topograph, there is a group of streaks whose curvature and trend are the same as the trace of the cutting on the back of the wafer. In order to determine the thickness of the surface damage layer of the polished wafer, some steps have been etched on the surface of the tested wafer with the controlled chemical step-etching technique. The height of the steps is of the order of 1 to 2  $\mu$ m. Fig. 2 shows the relation between the etching depths and the FWHMs for three polished GaAs wafers. The curves 1 and 2 show that the changes in the FWHMs tend to steady after the depths reach 1.4 and 3  $\mu$ m, respectively, but curve 3 may be considered a straight line. From the above result, we can draw the conclusion that the thicknesses of the damage layer are 1.4 and 3  $\mu$ m, respectively for wafer 1 and 2, but that there is no damage layer for wafer 3. From more experimental results, it is

shown that the surface damage layer on polished GaAs wafers is in general 1 to 4  $\mu$ m deep. The damage layer may result from cutting, lapping or polishing. It must be etched out before making a device, because it is very detrimental to the quality and the lifetime of a device.

Table 1. The FWHMs of some GaAs wafers before and after etching.

Number of the sample		1	2	3	4	5	6	7
FWHM	before etching	16.5	17.0	17.0	18.0	18.0	23.0	24.0
	(second of arc)	16.5	14.5	16.5	17.5	18.0	17.0	18.0



Fig. 1. X-ray double crystal reflection 224 reflection Cu K $\alpha_1$  40KV, 160mA. 1 st crystal, (111) Si 224 reflection

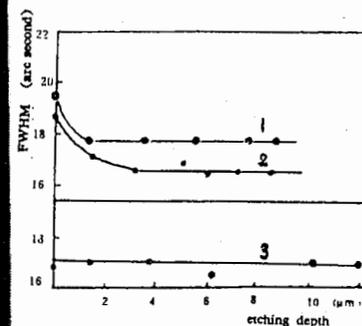


Fig. 2. Relations between etching depths and FWHMs of the (004) rocking curve.

**PS-13.03.13** DOUBLE CRYSTAL X-RAY ROCKING CURVE ANALYSIS OF MULTIPLE EPITAXIAL LAYERS. By Li Mei, Lin Chao Li, Zhang Zhi Shun and Ge Zhong Jiu, Changchun Institute of Physics, Academia Sinica, Changchun, 130021, China.

We have analysed the characteristics of GaAlAs/GaAs hetero-epitaxial structures by the X-ray double crystal rocking curve method. As an example, we have analysed a three-layer-epitaxial structure grown by LPE on a (001) GaAs substrate, where an active layer is sandwiched between two confining layers of GaAlAs (top layer and bottom layer, respectively). The active layer can be GaAs or a GaAlAs/GaAs (5\*5) superlattice layer. We have measured the rocking curve of a 400 reflection and observed the interference fringes for the above two kinds of samples. Using kinematical theory, we have calculated the lattice mismatch of the aluminium component. Computer simulation of the experimental curves have been performed with kinematical and dynamical diffraction, respectively. We have discussed the reason for the appearance of the interference fringes, and have calculated the thickness of the different layers. We can come to the following conclusions; 1. For sample No 1, we observed the diffraction peaks of the two confining layers and the substrate. There is a shoulder on the left side of the substrate peak. In the simulated curve, two types of interference fringes can be seen. One of them is located between the diffraction peaks, and the fringe period is 15 second of arc. It is related to the thickness of the top layer. The other type of fringes is superposed on the left side of the substrate diffraction peak. Its period of 7 second of arc corresponds to a thickness of 2.7  $\mu$ m. This is just the distance between the GaAs active layer and the substrate. This result implies that the interference fringes may be due to the interaction between the beams diffracted by the active layer and the substrate. 2. For sample No 2, the experimental and simulated curves are basically identical. There are fringes of period  $26 \pm 1$  seconds of arc. From the calculated result, it is determined that the interference fringes are not the Pendellösung fringes of the top layer. The interference fringes on both sides of the GaAlAs

peak are correlated with either the thickness of the top layer or that of the Al component. 3. For the two kinds of samples with different sandwiched layers, we calculated the aluminium component. The results calculated using the kinematical and dynamical theories were found to be different. The result obtained using dynamical theory is closer to the actual growth parameters because kinematical theory neglects the interaction between the beams diffracted by each layer. In fact, the above interaction will directly influence the shape and the position of the diffraction peaks.

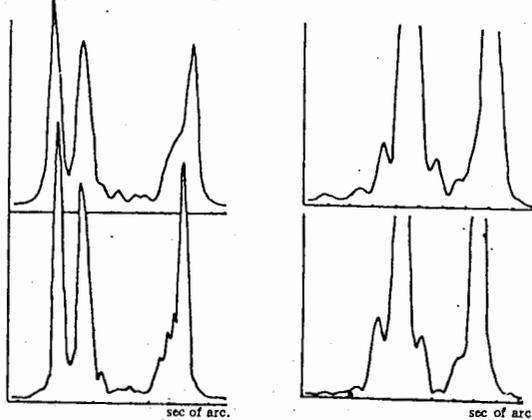


Fig. 1 Sample No. 1 (400) double crystal rocking curve. a) experimental curve; b) simulation curve.

Fig. 2 sample No. 2 (400) double crystal rocking curve. a) experimental curve; b) Best-fit simulation of the experimental curve;

**PS-13.03.14 MICROSTRUCTURAL CHARACTERISATION OF HEXAGONAL  $(\text{Ag,Cu})\text{Zn}_{14}$  ALLOYS IN THE DEFORMED AND AS-CAST STATE.** By Hiranmay Pal, Swapan Kumar Pradhan and Madhusudan De, Department of Materials Science, Indian Association for the Cultivation of Science, Jadavpur, Calcutta-700 032, INDIA

Zn-based ternary alloys containing Ag and Cu as solutes having characteristic hexagonal close-packed (hcp) structure  $(\text{Ag,Cu})\text{Zn}_{14}$  have been studied for microstructural evolution in the deformed and as-cast state in order to correlate the microstructure with mechanical property of the materials. X-ray diffraction (XRD) line profile analysis, optical microscopy and scanning electron microscopy (SEM) have been used to study the microstructure and the mechanical property studied through the measurement of microhardness. X-ray studies of the deformed alloys reveal increased influence of solute Cu in the generation of deformation stacking fault of density  $\alpha$  with increasing solute concentration. A reasonably good estimate of stacking fault energy parameter  $(\gamma/\mu)$  for the alloys has also been made. Increased influence of solute Cu has been observed to control the grain sizes and microhardness. A close correlation could be established between the microstructural parameters and microhardness signifying the interdependence of mechanical property with microstructure of the alloys.

**PS-13.03.15 THE SIMULATION OF BRAGG-CASE DOUBLE-CRYSTAL IMAGES OF DISLOCATIONS IN GAAS CRYSTALS** by W.K.Wierzchowski\*, Institute of Electronic Materials technology, ul. Wolczynska 133, 01-919 Warsaw, Poland

The numerical integration of Takagi-Taupin equations was applied to the simulation of back-reflection double-crystal topographic images of dislocations in GaAs crystals. The numerical program calculates the gradient of deformation field in the isotropic approximation taking into account the stress relaxation at the free surface. The thickness of the crystal slab is also assumed to be finite. In the present case where the absorption is very large the last assumption practically does not affect the simulated images and it allows to increase the size of the simulated image with a smaller computation time. It also enables placing the dislocation line further from the boundaries of the integration area. Taking into account the presence of epitaxial layer is also possible. In the present simulation, the finite divergence of the beam forming the image in the double-crystal method was taken into account. It was realized adding more than 60 images slightly differing in the assumed angle of incidence, weighted by appropriate rocking curves describing the angular distribution of radiation reflected by the monochromator.

The simulated images were compared to experimental images obtained with  $511\text{Ge}$ , -  $511\text{GaAs}$  double-crystal arrangement with  $\text{Cu K}\alpha 1$  radiation. Topographs were taken different equivalent reflections and the preliminary identification was on Lang transmission topographs.

A reasonable agreement between the simulated and experimental images was obtained, especially when a finite divergence of the beam was taken into account. In that case the interference fringes appearing close to the dislocation core in the plane-wave images were averaged.

The important feature of the simulated images was the presence of a characteristic black-white rosette. We proved a rough correspondence of it to the direct dilatation-orientation contrast coming from the displacement field. The contrast of the rosette was reversible with the change of the Burgers vector sign and with the change of the rocking curve slope. The extinction rules the reflection from equivalent crystallographic planes were significantly affected by surface stress relaxation phenomena.

Comparative simulation of dislocation images in crystals with lower absorption and smaller structure factors, such as silicon and diamond, exhibited a higher contribution of the different interference fringes at the expense of the rosette.